

Ab-initio simulation of phase change materials

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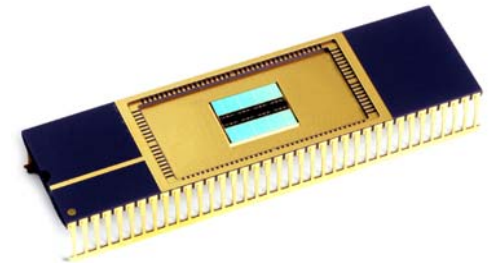
Projects: PHASEMAT (DECI-3) PHASEALL (DECI-4)

ETH Zurich c/o USI Lugano: S. Caravati, R. Mazzarello,
M. Parrinello (PI)

PSI, Villigen (CH): M. Krack

Phase Change Materials for electronic and optical data storage

Optical data storage: DVD



Electronic non-volatile memory: Phase Change Memory Cell

Chalcogenides alloys $\text{Ge}_2\text{Sb}_2\text{Te}_5$ (GST)

fast and reversible change between crystalline and amorphous phases (50 ns)

Phase change Materials

Two states system → possibility of storing a "0" or "1" bit

Large difference in properties between the two phases
(crystal - metallic amorphous - insulating)

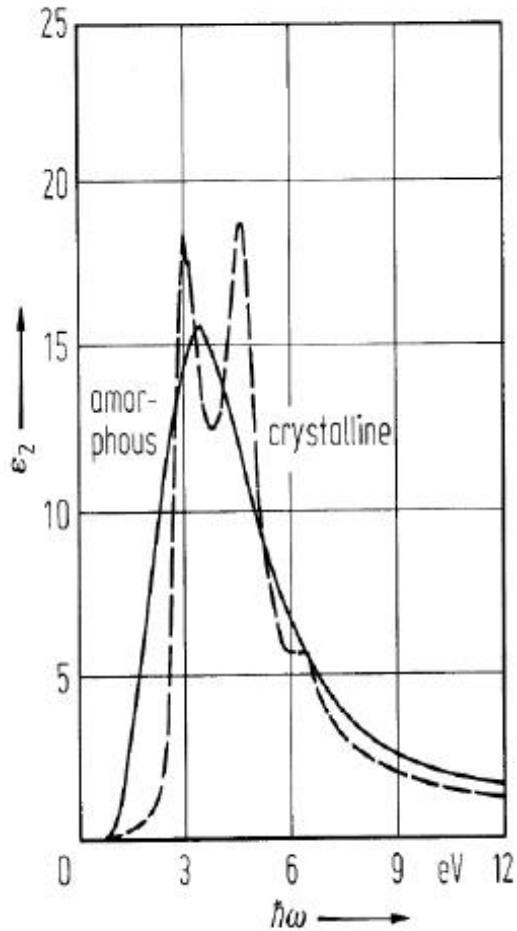
Resistivity changes by 3 orders of magnitude → PCM

Reflectivity change 30 %: → optical storage (DVD-RAM)

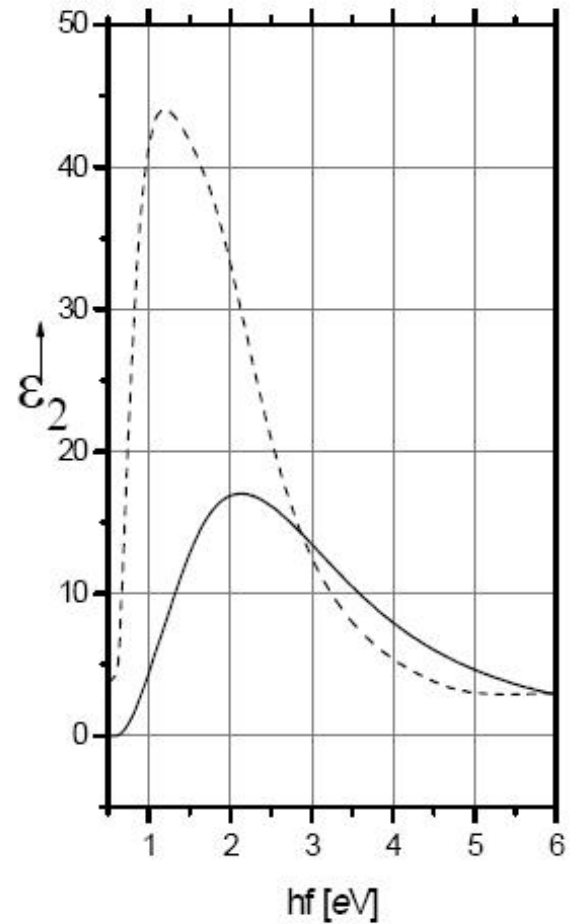
Transition induced by heating

Optical contrast between amorphous and crystal

GaAs



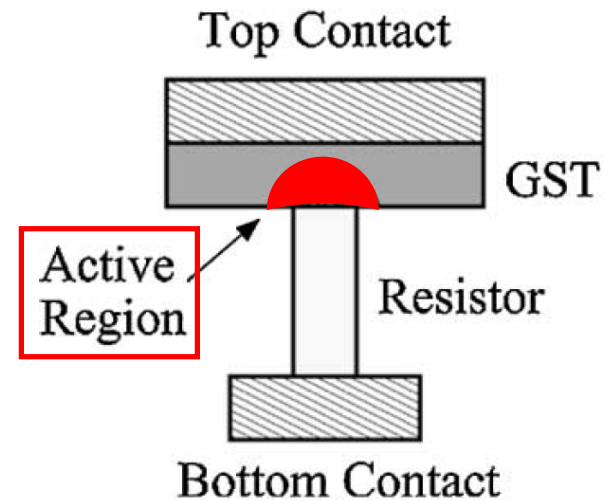
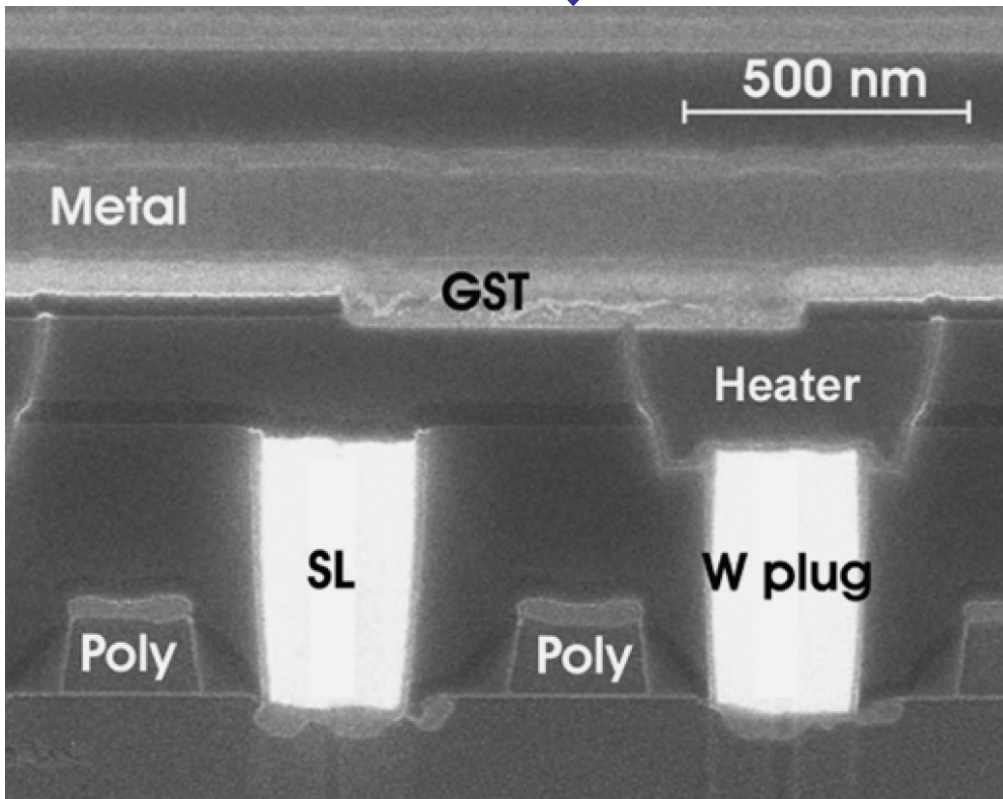
Ge₂Sb₂Te₅



The Phase-Change memory cell

Schematic representation

SEM cross section

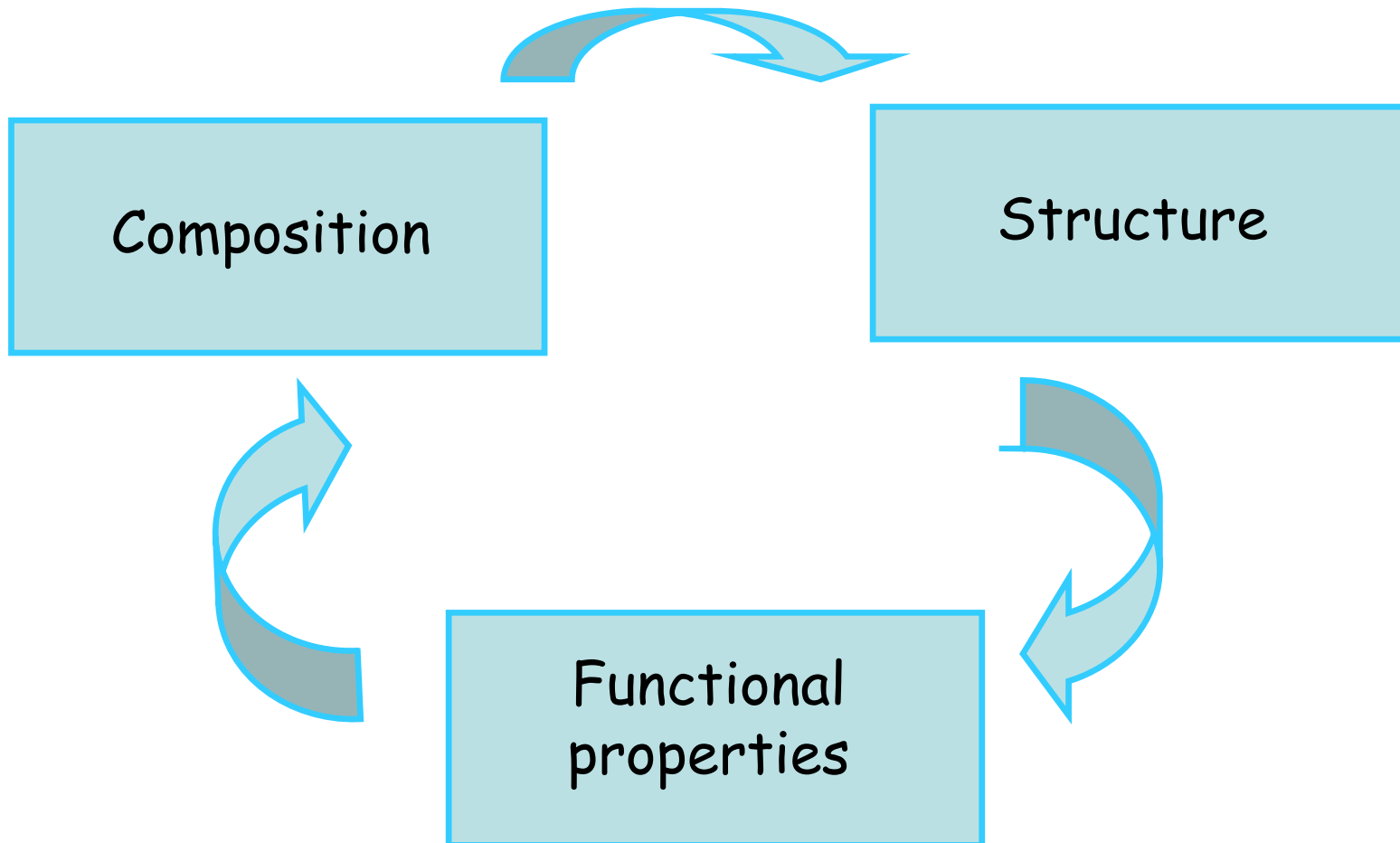


- Active region: a small drop within GST film undergoes the phase transition
- Phase-change by heating via Joule effect

Collaboration with Numonyx

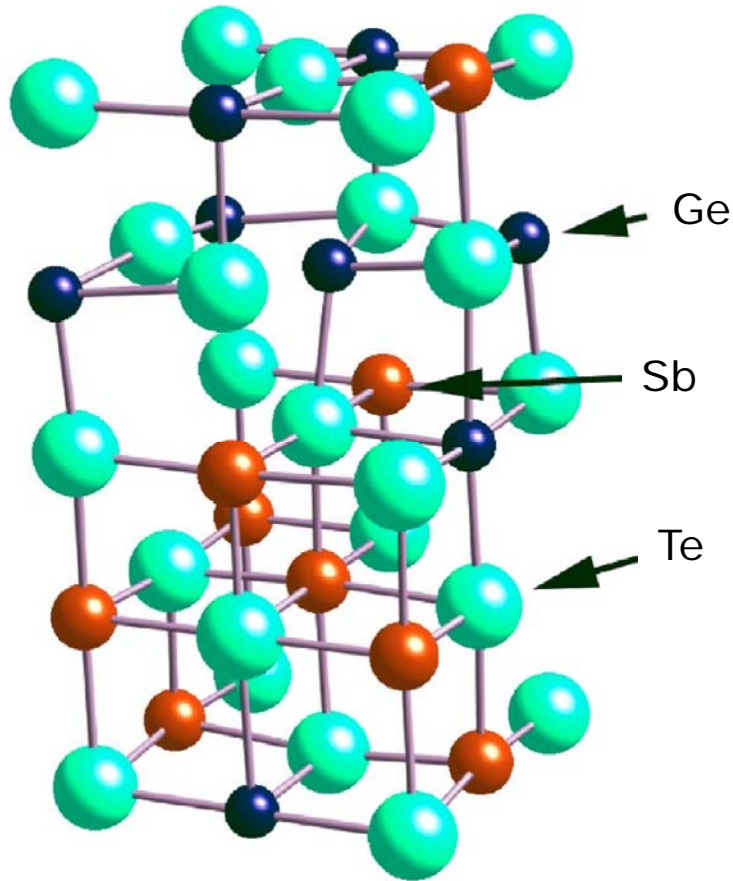
Needs for better performing materials for PCM

e.g. automotive applications require stability of amorphous at high temperature



GST crystalline phase: Rocksalt

❖ NaCl-type

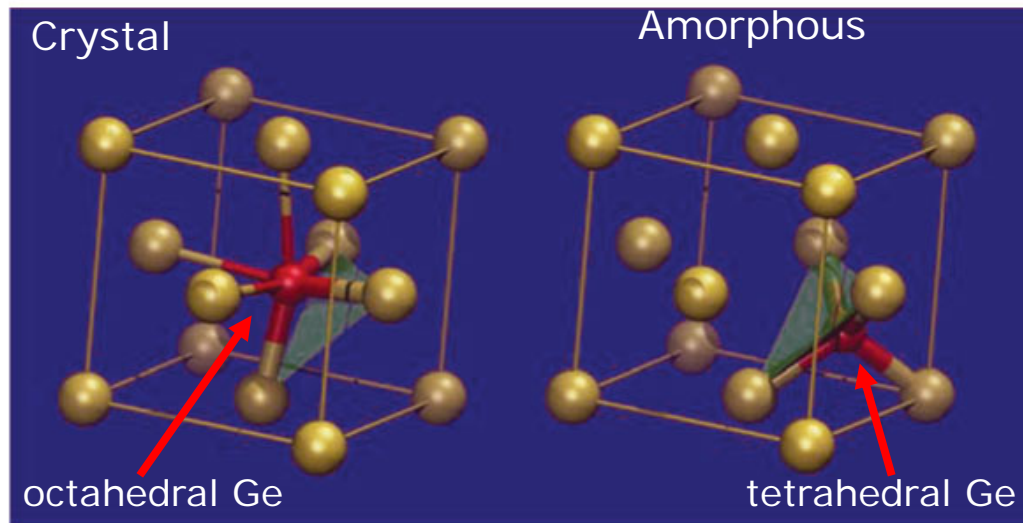


- One sublattice fully occupied by Te
- The other randomly occupied by Ge, Sb and 20% vacancies
- Octahedral environment each atom 6-fold coordinated

The Kolobov's picture of amorphous GST

- EXAFS/XANES data suggest 4-fold Ge coordination
- Ge in tetrahedral environment in amorphous GST
- Phase transition cubic \rightleftharpoons amorphous: octahedral \rightleftharpoons tetrahedral change of Ge environment

[Kolobov et al., *Nat. Mater.* **3** (2004)]



Octahedral \rightleftharpoons tetrahedral change responsible for electronic and optical contrast

Phase Change Materials from First Principles

- Ab-initio (Car-Parrinello) atomistic molecular dynamics simulations
- Density Functional Theory (GGA + pseudopotentials)

Second generation Car-Parrinello method (Kühne et al, PRL 2007)
Factor 25 speed up with respect to conventional ab-initio MD

- Models of amorphous phase (200-500 atoms) generated by quenching from the melt on 100 ps time scale
(1 ps \rightarrow 2 hours on 256 processors)
- Code Quickstep, open source CP2K project <http://cp2k.berlios.de>
Basis sets: Gaussian (wavefunctions) + plane waves (elec. density)
- Sites: Juelich (JSC), Garching (RGZ-MPG) Ibm sp6 Deci-3
Finland (CSC) Cray XT4/XT5 Deci-4

$\text{Ge}_2\text{Sb}_2\text{Te}_5$ (GST) GeTe Sb_2Te_3

InGeTe_2 (IGT) (high T_{crys} , Morikawa *et al*, IEEE 2007)

- Structural Properties
- Vibrational properties
- Optical properties (DVDs)
- Electronic properties (PCM)

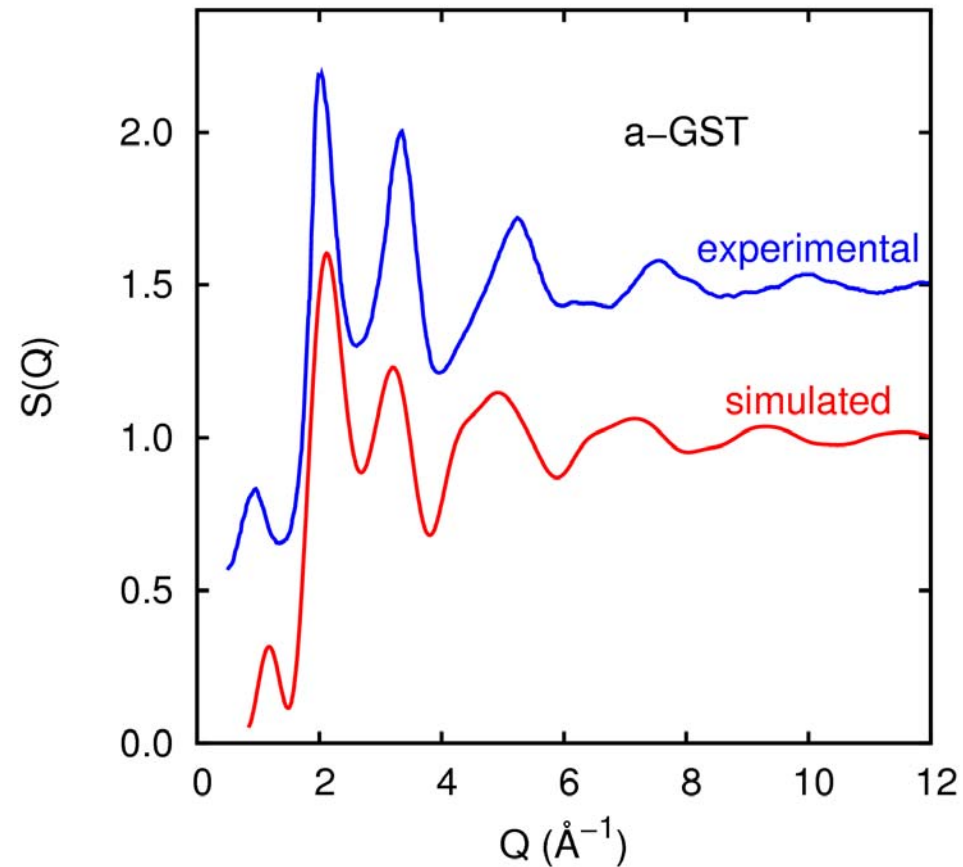
GST: amorphous model

Liquid equilibrated at 990 K
(270 atoms)

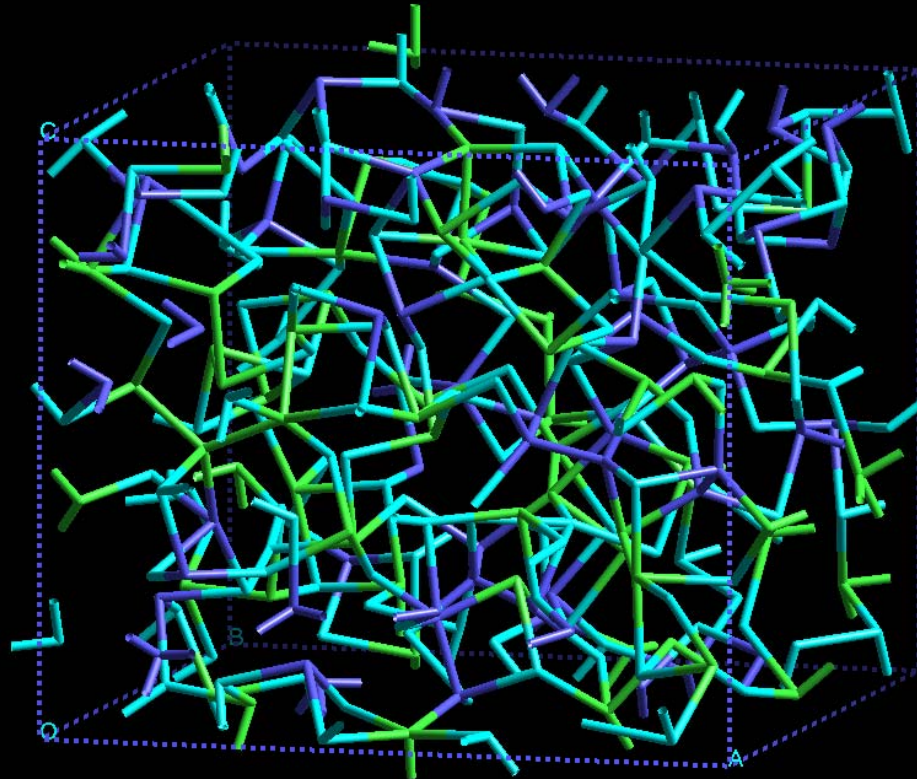
Liquid quenched to 300 K
(in 80 ps)

experimental neutron data:
Jovari et al., PRB 2007.

Structure Factor



GST: amorphous model



● Ge

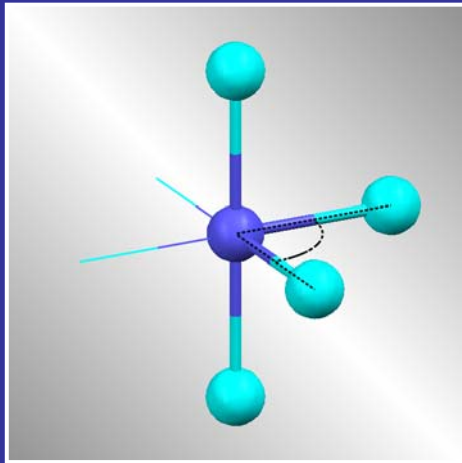
● Sb

● Te

Coexistence of octahedra and tetrahedra in amorphous PCM

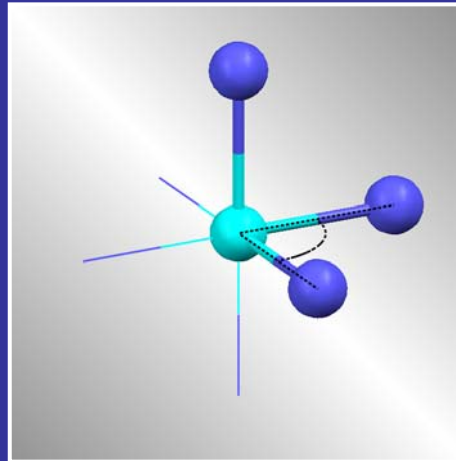
(defective) octahedral
Ge, Sb and Te

4-coor Ge

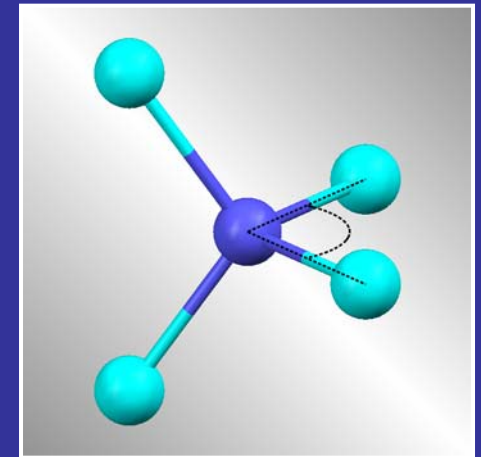


3+1 coordination
→ p-bonding

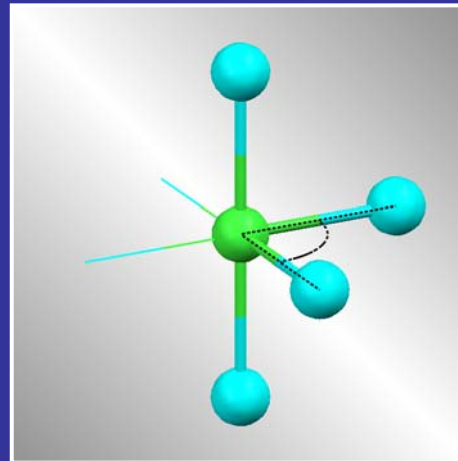
3-coor Te



tetrahedral
4-coor Ge

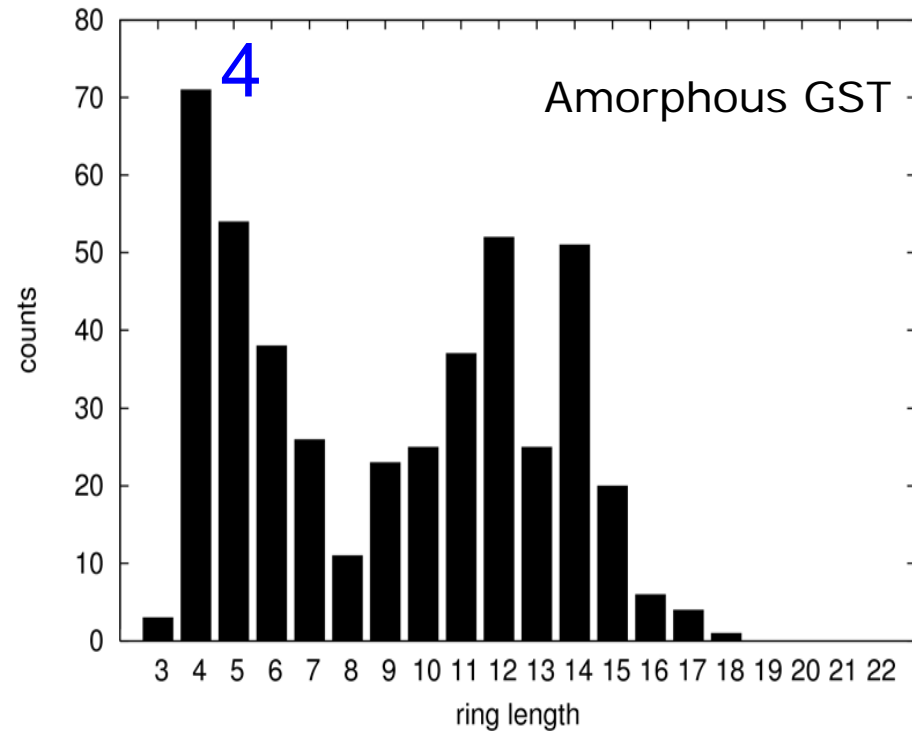


sp^3 bonding

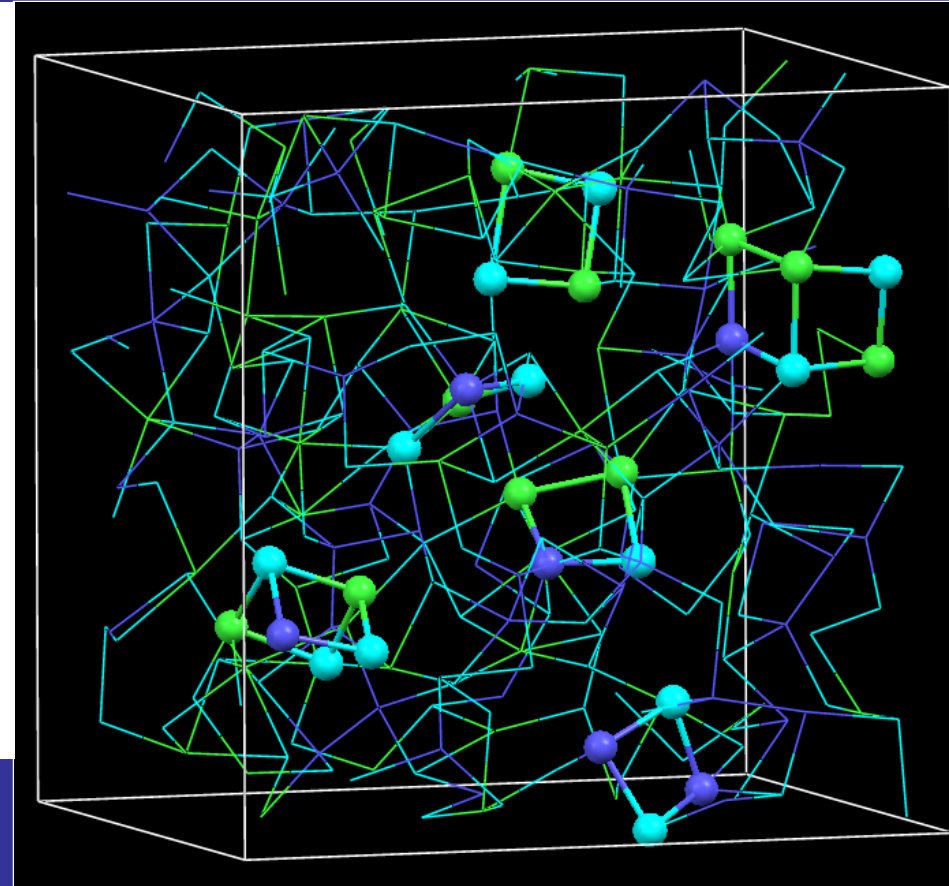


4-coor Sb

4-fold rings (squares)



Ring distribution



Caravati et al, APL 2007; J. Phys. Cond. Matt. 2009 ; PRL 2009; PRB 2010

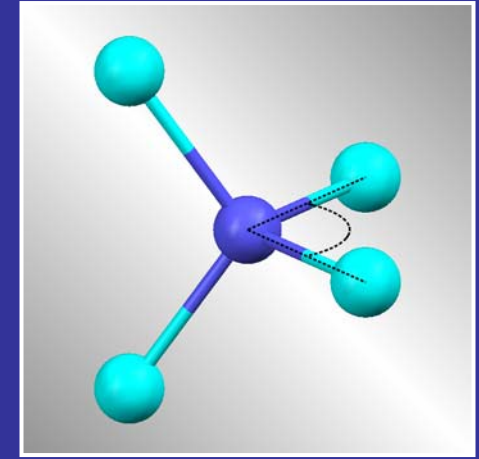
Akola and Jones PRB 2007; Hegedus and Elliott, Nat. Mater. 2008

Tetrahedral sites

*Ge-Ge and Ge-Sb bonds promote
Ge tetrahedral geometry*

*concentration of
tetrahedral Ge:*

- **GST: 27 % of Ge**
- **GeTe: 22 % of Ge**
- **InGeTe₂: 19 % of Ge**

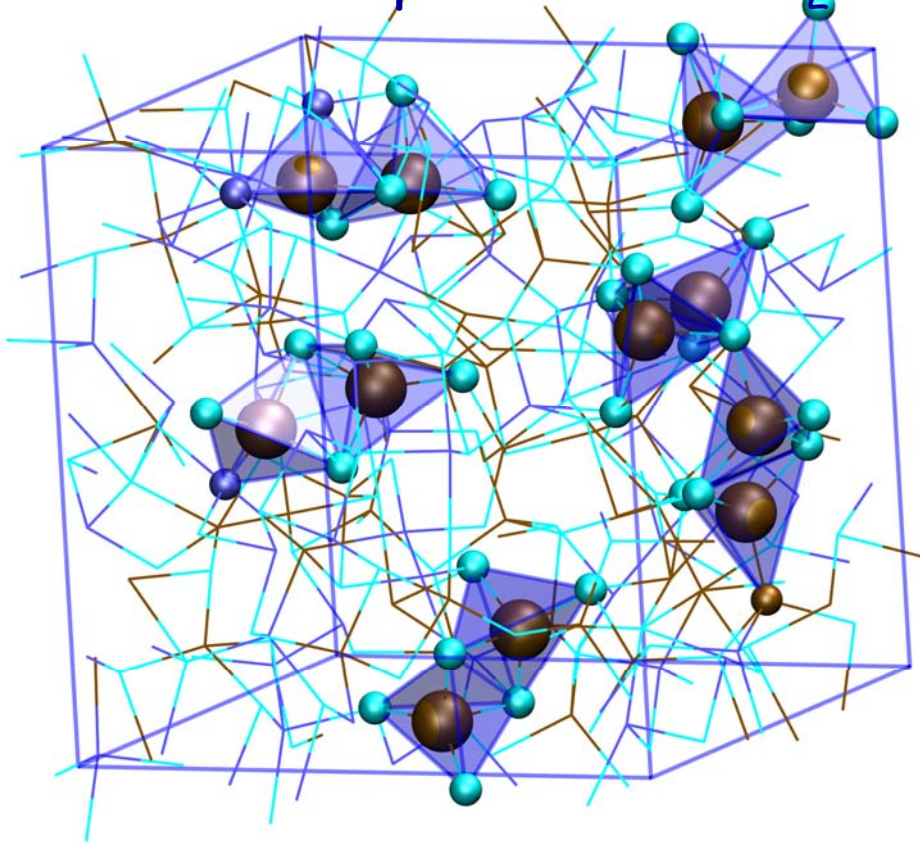


Sb₂Te₃ : only defective octahedral sites (No Ge)

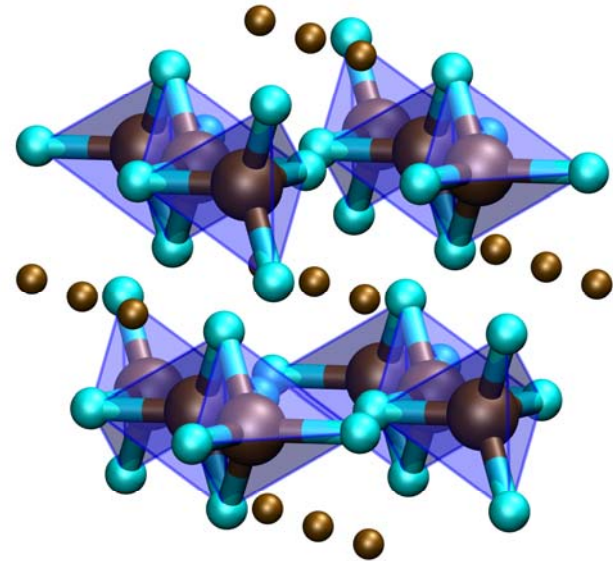
InGeTe₂ : In 60 % in tetrahedral sites !

InTe₄ tetrahedra in a-IGT

Amorphous InGeTe₂



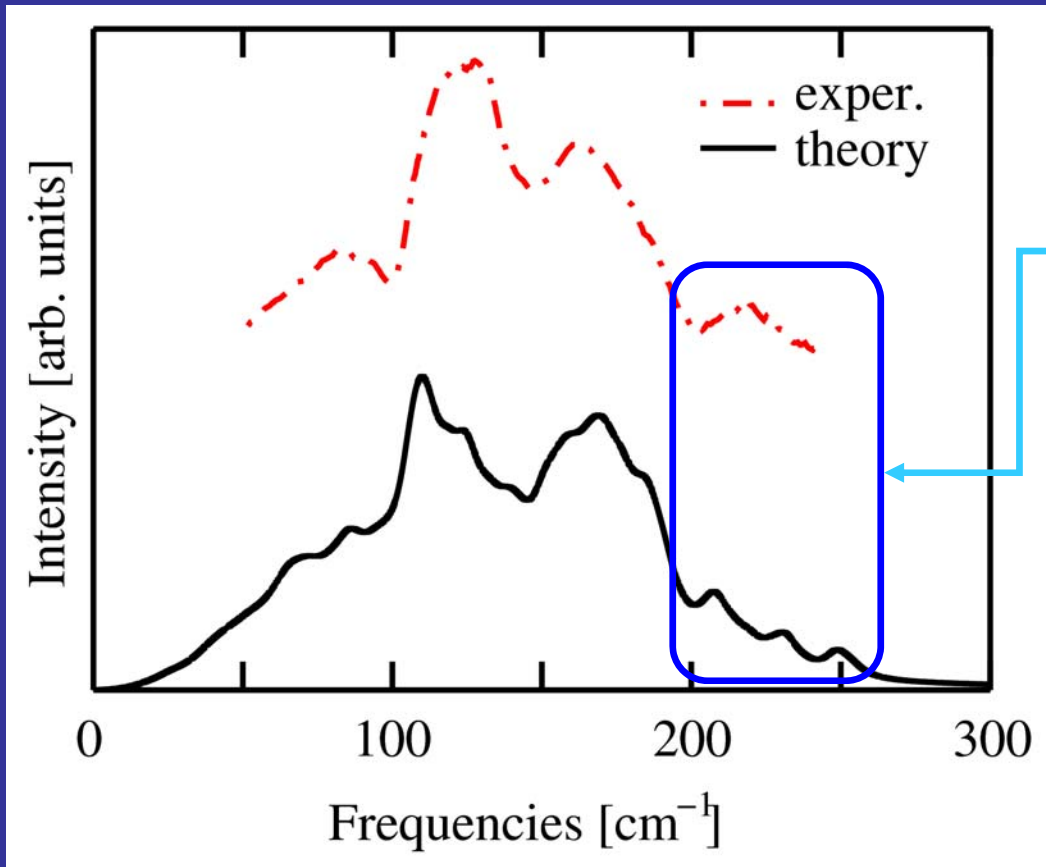
InTe crystal



Edge-sharing InTe₄ tetrahedra
in crystalline InTe and In₂Te₅

Tetrahedral Ge: vibrational fingerprint

Amorphous GeTe: Raman spectrum



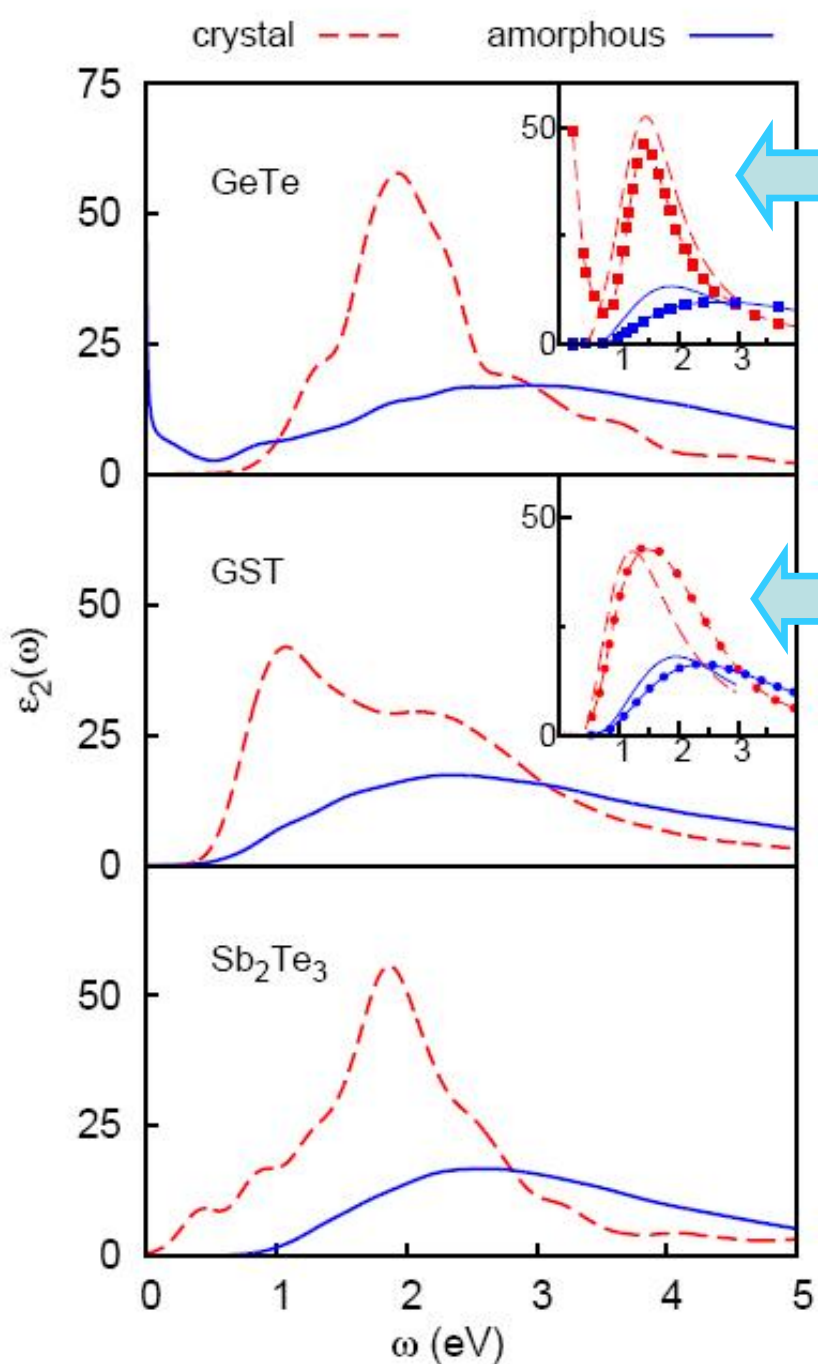
Phonon freq. > 190 cm⁻¹:

- Localized on tetrahedral Ge

⇒ signature of tetrahedral Ge

Mazzarello, Caravati, Angioletti, Bernasconi, Parrinello

Phys. Rev. Lett. 104, 085503 (2010)



Exp.

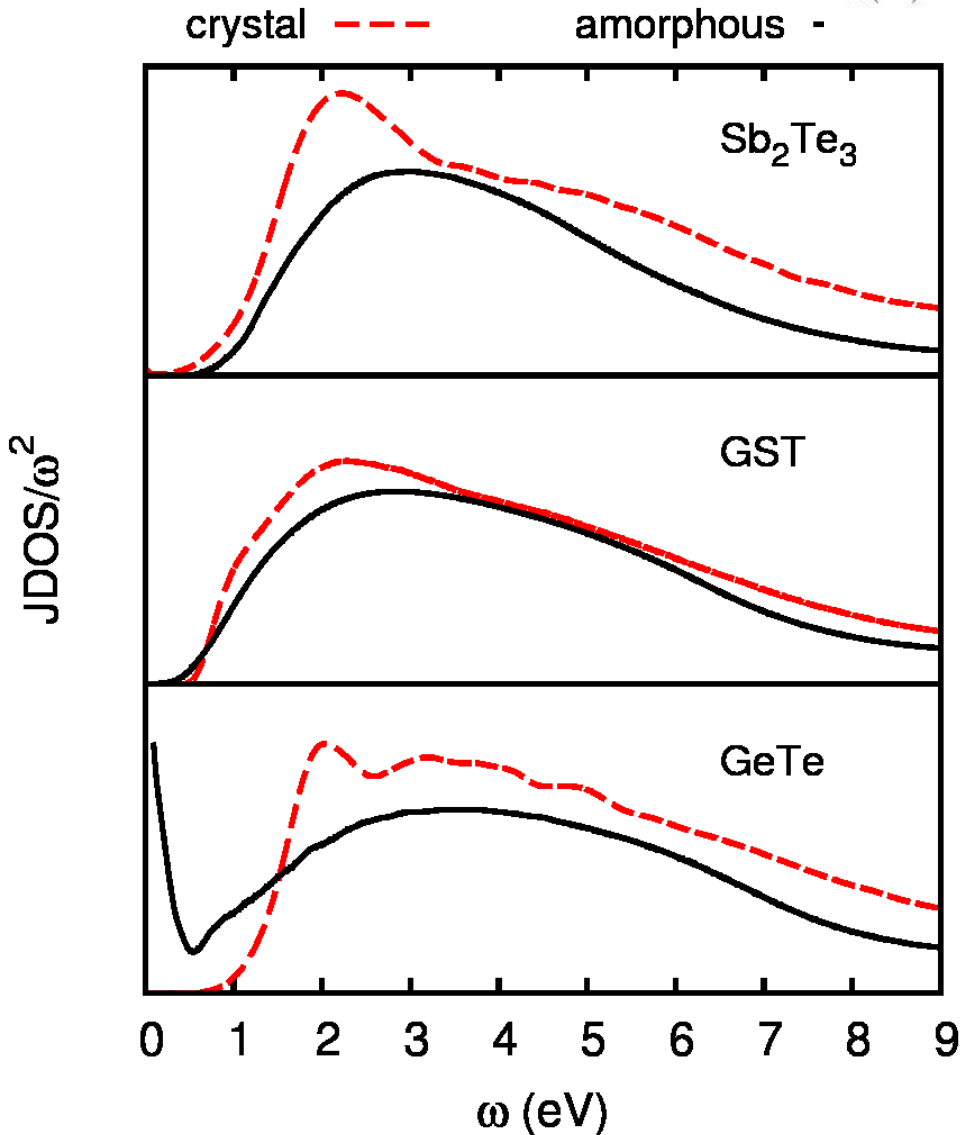
Optical contrast in PCM

Exp.

$\epsilon_2(\omega)$ dielectric functions
in RPA with hybrid
functional HS03

Joint Density of States

$$\epsilon_2(\omega) = \frac{8\pi^2}{3V_o N_{\mathbf{k}} \omega^2} \sum_{v,c,\mathbf{k}} |\langle c, \mathbf{k} | \mathbf{p} | v, \mathbf{k} \rangle|^2 \delta(\omega - E_{c,\mathbf{k}} + E_{v,\mathbf{k}})$$



(Huang and Robertson PRB 2010)

Angular disorder in p-bonds in amorphous phases



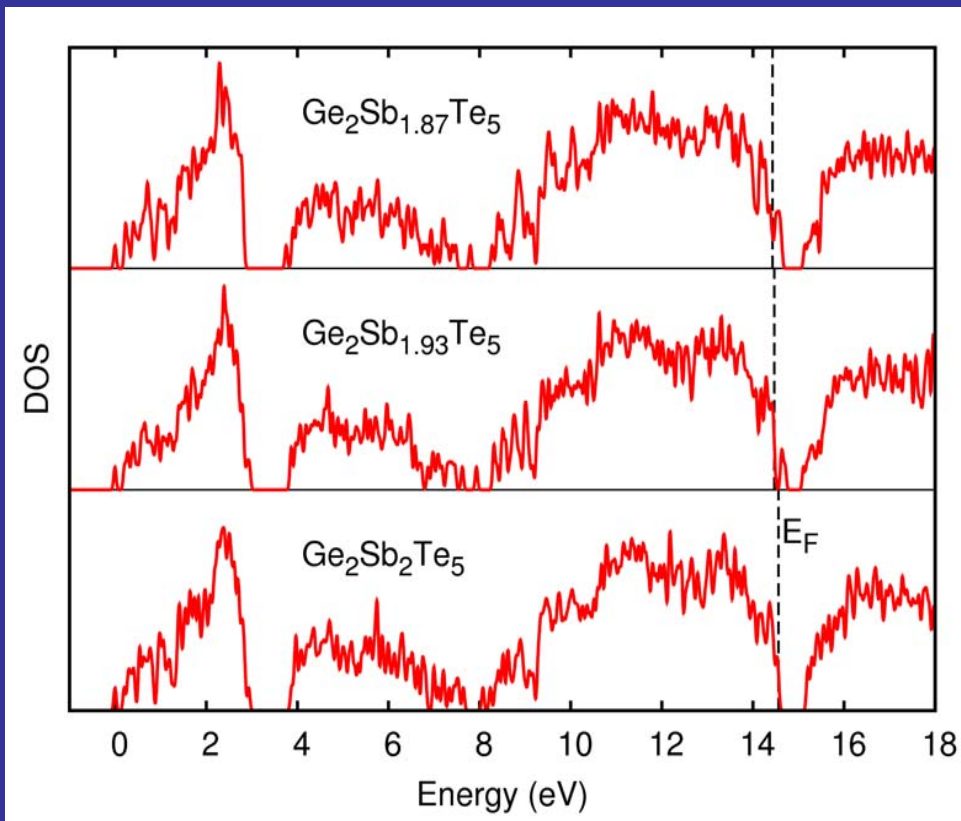
decrease of the optical matrix elements



optical contrast

Electronic properties

GST stoichiometric crystal is semiconducting



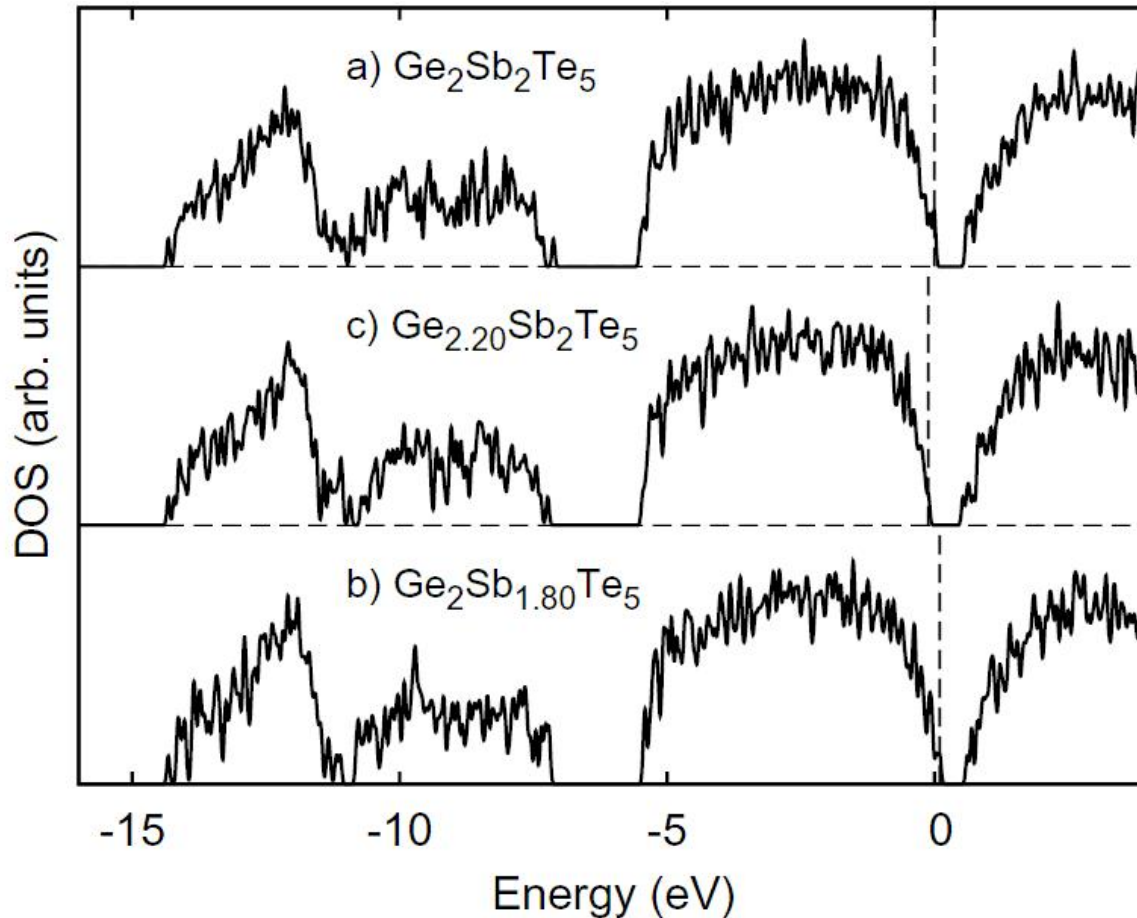
Ge/Sb deficiency: \rightarrow holes in VB

Ge/Sb excess: \rightarrow electrons in CB

HS03 hybrid functional

Electronic Properties

Non-stoichiometric GST: amorphous phase



Semiconductor:

Band gap ~ 0.6 eV

Ge excess, Sb deficiency \rightarrow No shift of E_f

Electronic properties

- Sb vacancies turn the cubic crystal into degenerate p-type semiconductor
- Electronic structure of amorphous phase is robust with respect to small change in composition

Conclusions

Ab-initio molecular dynamics simulations provided crucial insights on the properties of phase change materials

Structural properties of amorphous phases elucidated:
coexistence of defective octahedra and tetrahedra for Ge and In



Angular disorder in p-bonding as microscopic origin of optical contrast exploited in DVD

Difference in conductivity partially due to different response of the two phases to defects in stoichiometry

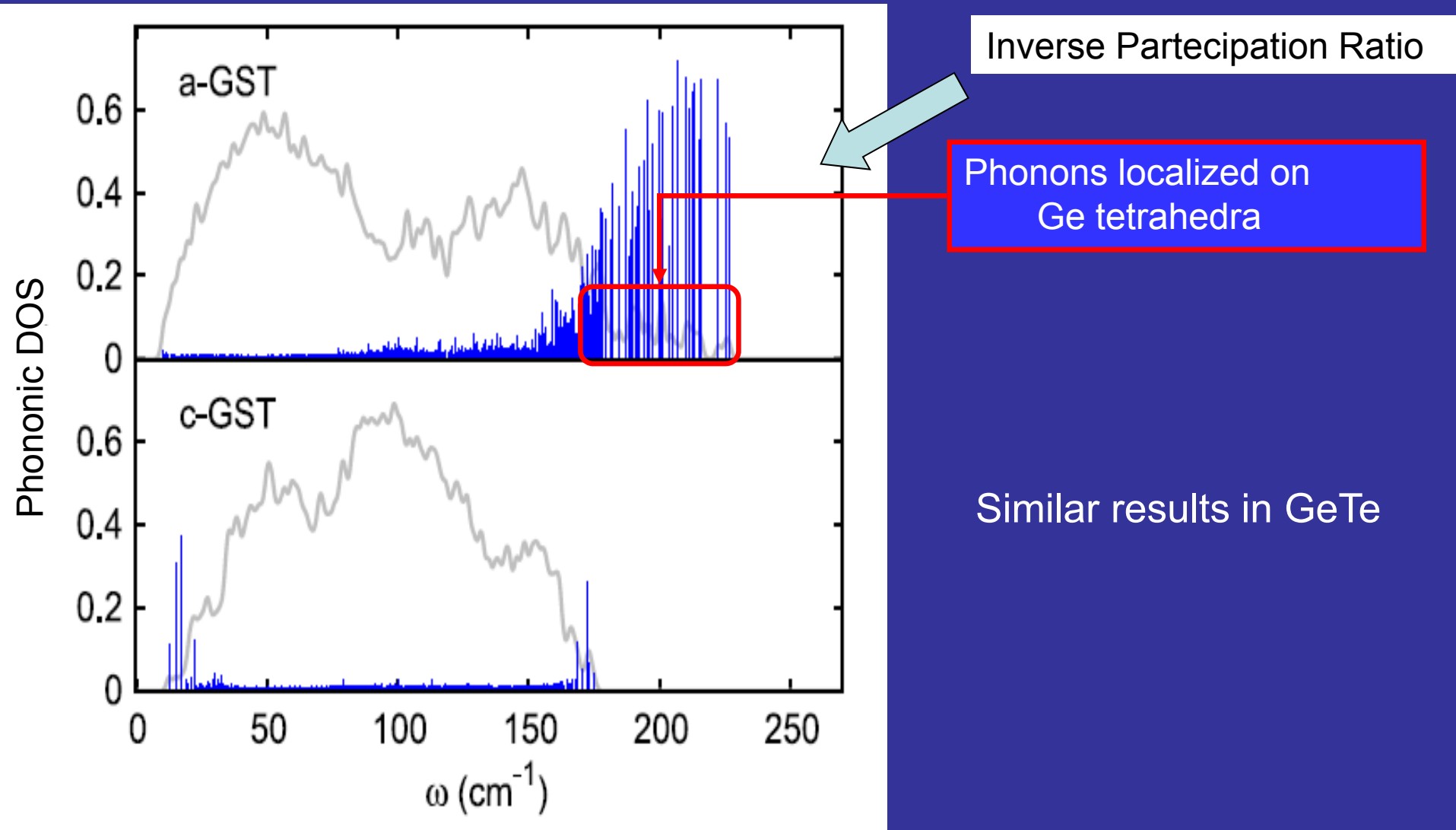
Aknowledgments

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CSCS, Manno (CH)

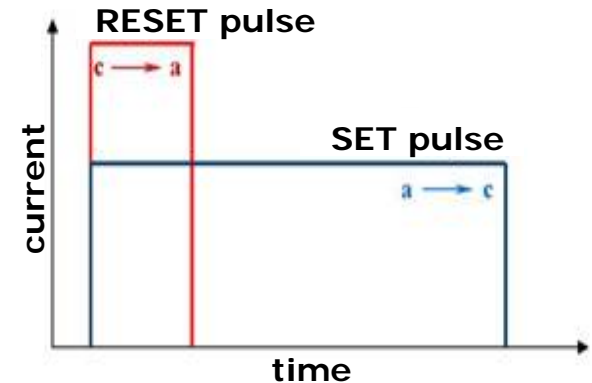
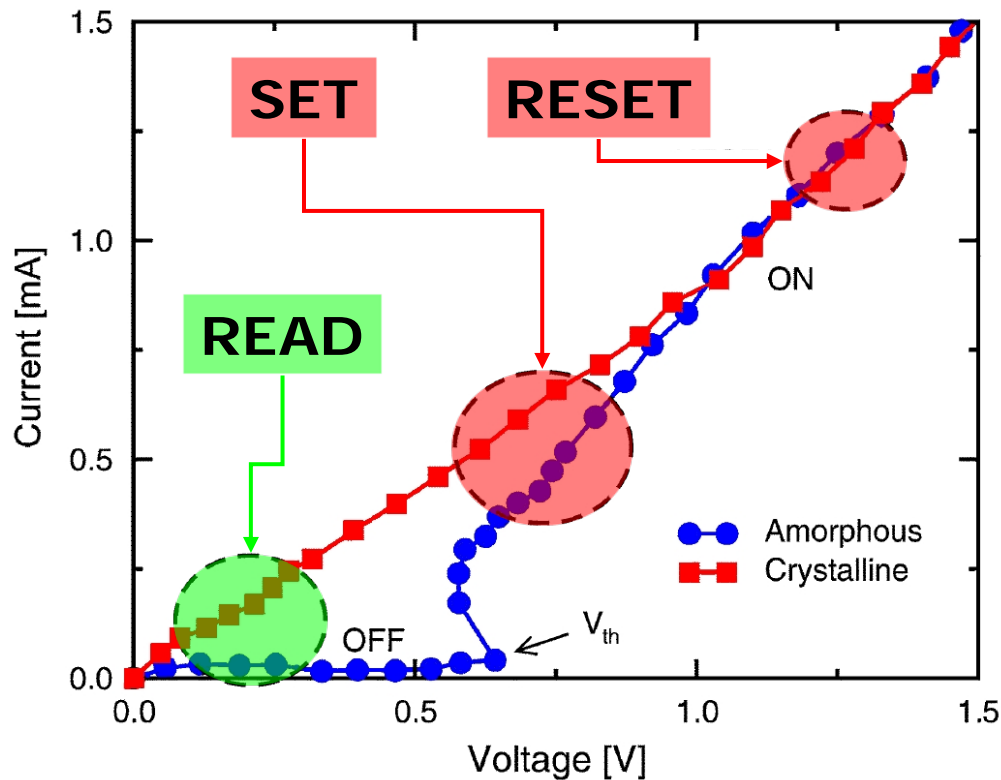
Thank you for attention !

Tetrahedral Ge: vibrational fingerprint



I(V) characteristic: threshold switching behavior

- cell readout: performed at low bias ($V < V_{th}$)
- set/reset: bias higher than threshold ($V > V_{th}$)



- RESET:
higher current &
shorter pulse
crystal → *amorphous*
- SET:
lower current &
longer pulse:
amorphous → *crystal*
DEISA - May 12, 2010